





Density-Functional-Theory Lecture II

Dr. Boris Kiefer
Physics Department
New Mexico State University

Overview: Tuesday

Review:

- Periodic table.
- Hartree-Fock: Exchange Interaction.
- Density-Functional-Theory (DFT).
- Exchange-correlation energy.

DFT: ins and outs:

- E_{\max}, K_{\max}
- Pseudopotentials vs. all-electron calculations.
- Metals and insulators.
- Convergence tests.
- Bulk/surface/molecules.
- Examples.

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Periodic Table

I												III		IV	V	VI	VII	VIII
H ¹ 1s _{1/2}																	He ² 1s _{1/2}	
Li ³ 2s _{1/2}	Be ⁴ 2s _{1/2}																B ⁵ 2p _{1/2}	
Na ¹¹ 3s _{1/2}	Mg ¹² 3s _{1/2}																Al ¹³ 3p _{1/2}	
K ¹⁹ 4s _{1/2}	Ca ²⁰ 4s _{1/2}	Sc ²¹ 3d _{3/2}	Ti ²² 3d _{5/2}	V ²³ 3d _{5/2}	Cr ²⁴ 3d _{5/2}	Mn ²⁵ 3d _{5/2}	Fe ²⁶ 3d _{5/2}	Co ²⁷ 3d _{5/2}	Ni ²⁸ 3d _{5/2}	Cu ²⁹ 3d _{5/2}	Zn ³⁰ 3d _{5/2}	Ga ³¹ 4p _{1/2}	Ge ³² 4p _{1/2}	As ³³ 4p _{1/2}	Se ³⁴ 4p _{1/2}	Br ³⁵ 4p _{1/2}	Kr ³⁶ 4p _{1/2}	
Rb ³⁷ 5s _{1/2}	Sr ³⁸ 5s _{1/2}	Y ³⁹ 4d _{3/2}	Zr ⁴⁰ 4d _{5/2}	Nb ⁴¹ 4d _{5/2}	Mo ⁴² 4d _{5/2}	Tc ⁴³ 4d _{5/2}	Ru ⁴⁴ 4d _{5/2}	Rh ⁴⁵ 4d _{5/2}	Pd ⁴⁶ 4d _{5/2}	Ag ⁴⁷ 5s _{1/2}	Cd ⁴⁸ 5s _{1/2}	In ⁴⁹ 5p _{1/2}	Sn ⁵⁰ 5p _{1/2}	Sb ⁵¹ 5p _{1/2}	Te ⁵² 5p _{1/2}	I ⁵³ 5p _{1/2}	Xe ⁵⁴ 5p _{1/2}	
Cs ⁵⁵ 6s _{1/2}	Ba ⁵⁶ 6s _{1/2}	71-79 Rare Earths	Hf ⁷² 5d _{3/2}	Ta ⁷³ 5d _{3/2}	W ⁷⁴ 5d _{3/2}	Re ⁷⁵ 5d _{3/2}	Os ⁷⁶ 5d _{3/2}	Ir ⁷⁷ 5d _{3/2}	Pt ⁷⁸ 5d _{3/2}	Au ⁷⁹ 6s _{1/2}	Hg ⁸⁰ 6s _{1/2}	Tl ⁸¹ 6p _{1/2}	Pb ⁸² 6p _{1/2}	Bi ⁸³ 6p _{1/2}	Po ⁸⁴ 6p _{1/2}	At ⁸⁵ 6p _{1/2}	Rn ⁸⁶ 6p _{1/2}	
Fr ⁸⁷ 7s _{1/2}	Ra ⁸⁸ 7s _{1/2}	89-103 Actinides	Rf ¹⁰⁴ 5f ¹⁴ 6d ²	Ha ¹⁰⁵														
Rare earths (Lanthanides)		La ⁵⁷ 5d ¹	Ce ⁵⁸ 4f ¹	Pu ⁵⁹ 4f ¹⁴	Nd ⁶⁰ 4f ⁴	Pm ⁶¹ 4f ⁷	Sm ⁶² 4f ⁶	Eu ⁶³ 4f ⁷	Gd ⁶⁴ 4f ⁷	Tb ⁶⁵ 4f ⁹	Dy ⁶⁶ 4f ¹⁰	Ho ⁶⁷ 4f ¹¹	Er ⁶⁸ 4f ¹²	Tm ⁶⁹ 4f ¹³	Yb ⁷⁰ 4f ¹⁴	Lu ⁷¹ 5d ¹ 4f ¹⁴		
Actinides		Ac ⁸⁹ 5d ¹	Th ⁹⁰ 6d ²	Pa ⁹¹ 5f ²	U ⁹² 5f ³	Np ⁹³ 5f ⁴	Pu ⁹⁴ 5f ⁶	Am ⁹⁵ 5f ⁷	Cm ⁹⁶ 5f ⁷	Bk ⁹⁷ 6d ¹ 5f ⁷	Cf ⁹⁸ 5f ¹⁰	Es ⁹⁹ 5f ¹¹	Fm ¹⁰⁰ 5f ¹⁴	Mg ¹⁰¹ 5f ¹⁴	No ¹⁰² 5f ¹⁴	Lr ¹⁰³ 5d ¹ 5f ¹⁴ 6d ¹		

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Adiabatic Decoupling

Decouple electronic and nuclear degrees of freedom:

- ➔ Nuclei: classic treatment.
- ➔ Electrons: quantum mechanics.

If T=0 K ➔ Kinetic energy of the nuclei is zero.
Note: Nuclei are **not** eliminated from the problem, $U_{\text{nuc-nuc}}$ and $U_{\text{nuc-el}}$.
 Static problem.

If T>0 K ➔ Nuclei are no longer fixed. Lattice vibrations; thermodynamics.

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Kohn and Sham Equations

Kohn and Sham (1965)

$$\left[-\frac{\hbar^2}{2m_{el}} \nabla_r^2 + V_{eff}(r, n(r)) \right] \phi_i(r) = \varepsilon_i \phi_i(r)$$

$$V_{eff}(r, n(r)) = V(r) \quad \text{External potential.}$$

$$+ e^2 \int \frac{n(r')}{|r-r'|} dr' \quad \text{Coulomb interaction.}$$

$$+ \frac{\delta E_{xc}(n(r))}{\delta n(r)} \quad \text{Exchange-correlation.}$$

Can be generalized to include magnetism:

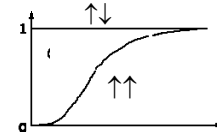
$$n = n_{\uparrow} + n_{\downarrow}$$

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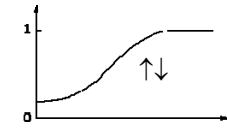
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What is the Exchange-Correlation Potential, E_{xc} ?

Thermodynamics: ground state == lowest energy state



Pauli exclusion principle
→ Exchange energy.



Repulsion of opposite spins
→ Correlation energy.

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Exchange Correlation Functionals

Active research field to develop better E_{xc} functionals:

- LDA : $E_{xc} = E_{xc}(\rho)$
Ceperley and Alder (1981)...

- GGA : $E_{xc} = E_{xc}(\rho, \nabla \rho)$
PW91 (1991), PBE (1996),
revPBE (1996), rPBE (1999)...

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Exchange Correlation Functionals: Which one to choose?

Depends on system and variable of interest:

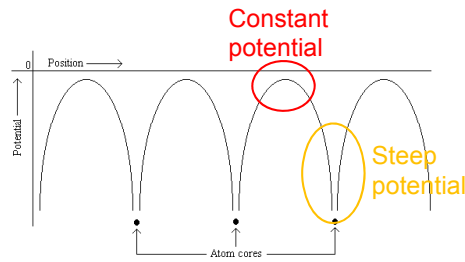
Iron: LDA gives *incorrect* ground state (hcp rather than bcc); GGA *correct* ground state.

SiO₂: LDA gives better agreement with experiment at high pressures than GGA.

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Suitable Electronic Wave Functions



Solutions to Schrödinger equation with **constant potential**:

Planewaves: $\psi \sim \exp(i k r)$
 Works best for "flat" potentials.

<http://gauss.adsmt.edu/~bhemmel/quantu1.gf>

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Brillouin zones

Crystal: Periodic Structure in 3-d space

Planewaves: $\psi \sim \exp(ikr)$

→ Fourier transform.

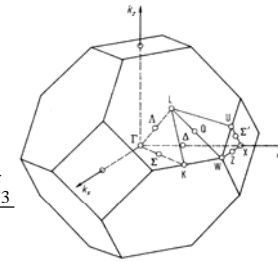
1st Brillouin zone: range of independent values.

Definition:

$$\vec{k}_i = \frac{n_i}{N_i} \vec{b}_i; \quad \vec{b}_1 = 2\pi \frac{\vec{a}_2 \times \vec{a}_3}{V_{cell}}$$

1,2,3 cyclic

Same as in xrd.



Face-centered cubic

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Energy Cutoff

Planewaves: $\psi \sim \exp(ikr)$

Energy:

$$E(\vec{k}) = \frac{\hbar^2}{2 \cdot m_e} \vec{k}^2$$

Define cutoff energy: E_{max}

$$E_{max} = \frac{\hbar^2}{2 \cdot m_e} \vec{k}_{max}^2$$

Note:

$$k = \frac{2\pi}{\lambda} \Rightarrow E_{max} \uparrow \rightarrow k_{max} \uparrow \rightarrow \lambda_{min} \downarrow$$

Increase E_{max} to resolve finer details of the charge density.

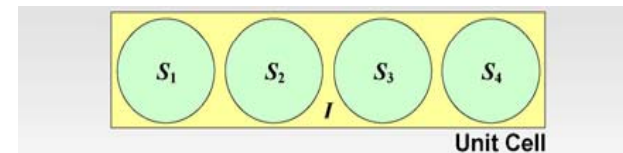
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Pseudopotentials

"Problem": Steep potential close to nuclei

→ plane waves are not well suited.



Idea: Most material/chemical properties rely on valence electrons.



Eliminate core electrons

Pseudopotential.

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Generation of Pseudopotentials

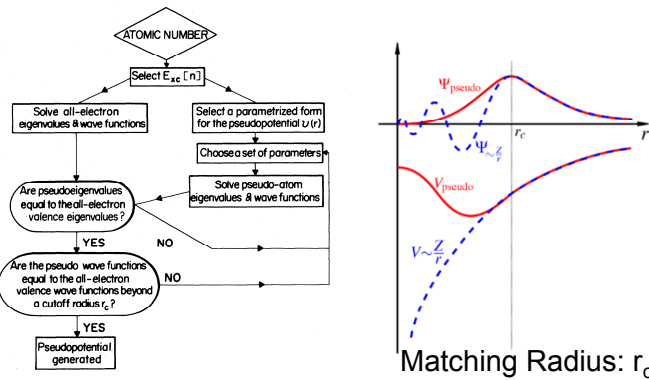


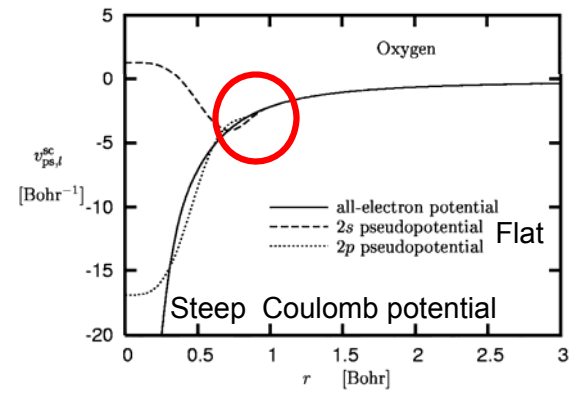
FIG. 6. Flow chart describing the construction of an ionic pseudopotential for an atom.

<http://upload.wikimedia.org/wikipedia/en/thumb/7/77/e/SketchPseudopotentials.png/300px/SketchPseudopotentials.png>

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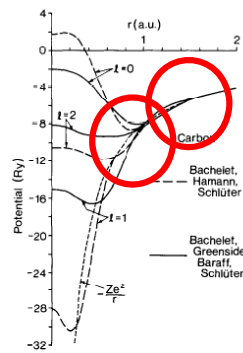
Oxygen: an example



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Carbon



Smaller r_c :
 → Better transferability.
 → higher energy cutoff.

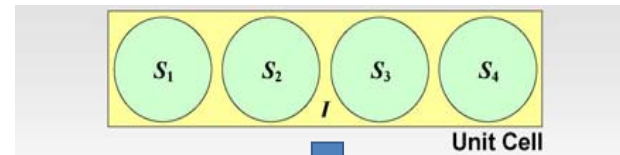
Pickell, 1989

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Alternative

Separate space in different regions:
 Interstitial regions → plane waves as before.
 Regions around the nucleus:
 use atomic-like states.



All-electron methods.
 (structural optimization difficult to implement, i.e. WIEN2k)

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Computational Approaches

Plane waves

Unit Cell

Interstitial Space: Plane Waves
Treatment of Space Close to Nuclei

Plane waves Atomic states

Pseudo-Potential Method	All Electron Method
VASP	Wien2K
Siesta	Fleur
PWScf	LAPW
Abinit	
	CASTEP

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Computational Procedure

```

    graph TD
      A[Construct V_ion given atomic numbers and positions of ions] --> B[Pick a cutoff for the plane-wave basis set e^{i(k+G)·r}]
      B --> C[Pick a trial density n(r)]
      C --> D[Calculate V_H(n) and V_XC(n)]
      D --> E[Solve Hψ = [-ħ²∇²/2m + V_ion + V_H + V_XC]ψ = εψ by diagonalization of H_{k+g, k+g'}]
      E --> F[Calculate new n(r)]
      F --> G{IS SOLUTION SELF-CONSISTENT?}
      G -- YES --> H[Compute Total Energy]
      G -- NO --> I[Generate New Density n(r)]
      I --> C
  
```

FIG. 7. Flow chart describing the computational procedure for the calculation of the total energy of a solid, using conventional matrix diagonalization.

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Types of Materials Density of States

Metal: States available just above the highest occupied state

Insulator: NO states available just above the highest occupied state

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Metals

Fermi-Dirac Distribution Function

Fermi-Dirac distribution for several temperatures

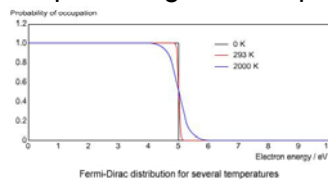
Metal: Occupation is a step-function.
"Softens" as temperature increases.

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Metals: Consequences - I

Problem:

At $T=0$ K (low temperatures)
Rapid change of occupation from 1 to zero.



During optimization: unoccupied wave functions become occupied and vice versa
→ Oscillations and instabilities.

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Metals: Consequences-II

Need to know charge density (wave functions) in the (1st) Brillouin zone very well

→ Expectation that we need a dense k-point mesh.

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Convergence Test

Carefully check variation of
 $E=E(E_{\max}, k\text{-point mesh})$

Two steps:

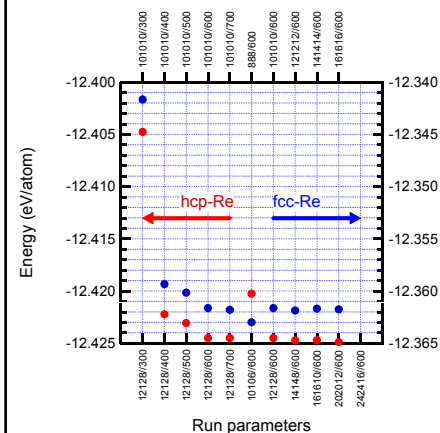
- 1) Fix E_{\max} and increase k-point mesh size.
- 2) Fix k-point mesh size and vary E_{\max} .

Changes between subsequent calculations should be less than 1 meV/atom.
(criterion may vary for specific applications)

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Convergence Test: Rhenium



Rhenium (Re):

- Convergence: hcp-Re: 600 eV; 12x12x8.
- fcc-Re: 600 eV; 10x10x10.
- $E(\text{hcp}) < E(\text{fcc})$
→ hcp-Re more stable, as observed experimentally.
- (Not shown: Pt: 600 eV; 14x14x14).

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Insulators

Occupied and unoccupied states are separated by an energy gap of finite width.

→ We do not expect instabilities and oscillations due to changing occupation numbers.

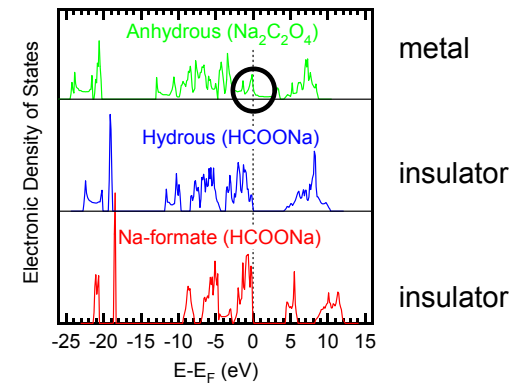
→ Less dense k-point meshes than for metals.

Even though not shown here experience shows that this expectation is supported.

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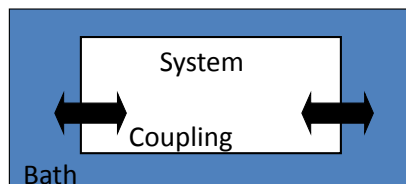
Example: Computed Density of States



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Temperature: Extended System Approach (Anderson, 1980)



Idea: 0th law of thermodynamics: Bring system in contact with heat bath.

Thermostat: $\langle T \rangle = \text{constant}$.

→ Choose coupling such that the proper ensemble is modeled (Nose, 1984a, 1984b).

But other options exist: Lattice-dynamics.

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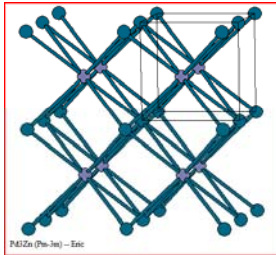
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Lattice Dynamics

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Bulk-Structures-I



Hypothetical:
Cubic-Pd₈Zn₈
Cubic-Pd₅₀Zn₅₀

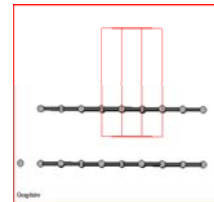
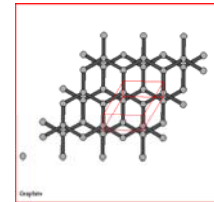
Bulk-calculations:

- Periodic → good match with plane waves.
- Solid solutions (alloying): Vary composition. Pd₈Zn₈ → Pd₉Zn₇ ...
- Disorder: fixed composition. Exchange Pd ↔ Zn.

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Bulk-Structures-II



Graphite:

- sp² – hybridization.
- intra-plane sigma bonds.
- inter-plane van der Waals. Origin: fluctuating dipoles.

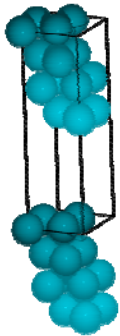
But:

- theory here: time independent → van der Waals compounds not necessarily well described. (New functionals).

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Surfaces-I



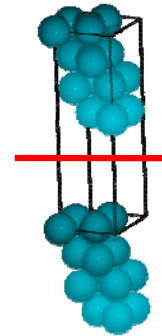
fcc-Pd(111)-slab.

- 2-d periodic.
 - Symmetry broken in 3rd dimension.
 - Vacuum layer (10-20 Å).
 - No electrons in vacuum. But: we use planewaves... → Need higher energy cutoff to make charge density zero in vacuum.
- Alternative: do not use planewaves. But more localized functions: wavelets, finite-elements...

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Surfaces-II



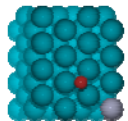
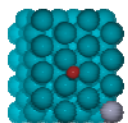
fcc-Pd(111)-slab.

- slab thick enough such that surface relaxation can be accounted for. 4 layers sufficient, relax top two layers (test convergence).
- asymmetric slab: *Artificial* induced (electrical) dipole moment. Long ranged → two options:
 - 1) make vacuum thick enough, 15 – 20 Å.
 - 2) **Compensate by additional dipole. (VASP: middle of the cell).**

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Surfaces-III



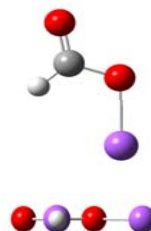
fcc-Pd(111)-slab.
(top-view)

- adsorption.
Example: CO, MeOH, EtOH,...
- Many molecules are polar ($\Delta EN \neq 0$)
- Same problem as with the asymmetric slab, same options to solve the problem.

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Molecules



- Electrons are localized.
- Same difficulty as for surfaces.

Note: Adsorption energies are energy differences
→ At least partial compensation.

Alternatives to plane waves are:
Wavelets and finite elements.

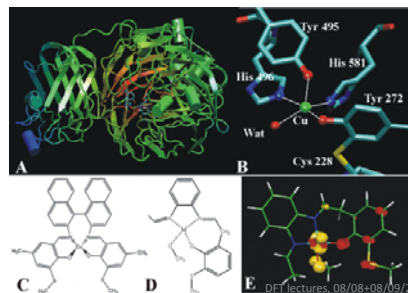
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Quantum Mechanics + Molecular Mechanics/dynamics

Hybrid methods: QM/MM/MD

- “Local” → quantum mechanics.
- “Non-local” → classical mechanics/molecular dynamics.

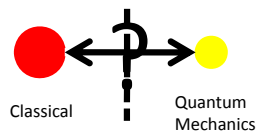


Advantage:

- Focus on active portion.
- Large scale; details are “less” important.

Disadvantage:

- Boundary: QM + MM?
- “Cutting” through bonds.



<http://bio.phys.uniroma1.it/gosse.gif>

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Some Applications of DFT

Earth's and Planetary Sciences:

- Electrons in planetary interiors.
- Melts at high pressure.

Environmental Geosciences:

- Structure of $(\text{H}_3\text{O})^+\text{Fe}(\text{SO}_4)_2$

Student presentations

- Andrew: $\text{Pt}_{1-x}\text{Re}_x$ solid solutions.
- Eric: Pd on $\gamma\text{-Al}_2\text{O}_3$.
- Levi: Pd on $\alpha\text{-Al}_2\text{O}_3$.
- Sam: non-Pt based catalysts.

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Planetary Interiors

(Stixrude et al., 1998)

ULTRAHIGH-PRESSURE MINERALOGY

Density (Mg m^{-3})

Charge Density (Atom/m)

Plasma model

$$P = 0.176 r_s^5 \cdot [1 - (0.407 \cdot Z^{2/3} + 0.207) \cdot r_s]$$

$$r_s = \left(\frac{3}{4\pi\rho} \right)^{1/3}$$

Gellman and Brueckner (1949)
Hubbard (1984)

Explanation:

- Coulomb attraction important.
- Electrons are not a plasma.
- Predicted density is too HIGH.
- Electronic screening.

Observations:
Charge density too high.

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Melts at High Pressure

(Mookherjee et al., 2008)

Background:

- Constitution of the Earth's interior.
- Abundance and distribution of hydrogen.
- Hydrogen affects: density, viscosity, elasticity, electrical conductivity, ...

12 * MgSiO₃ + 8 * H₂O = 84 atoms. 0 – 120 GPa, 3000 – 6000 K.

1 GPa ~ 10⁴ atm

T=3000 K

Low pressure:
OH⁻, H₂O, Mg-OH₂.

T=3000 K

High pressure:
also Si-O-H₂O-O-Si

Pressure dependent H₂O speciation:

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Implications

Figure 3 | Self-diffusion coefficient of hydrogen. Results at 3,000 K (blue), 4,000 K (green) and 6,000 K (red) from our simulations (symbols) and an Arrhenius fit to the simulation results (lines) (equation (1) with $D_0 = 9.2 \times 10^{-7} \text{ m}^2 \text{ s}^{-1}$, $E^* = 86 \text{ kJ mol}^{-1}$ and $V^* = 0.11 \text{ cm}^3 \text{ mol}^{-1}$). The value of the activation energy is similar to that found in hydrous basaltic melts at low pressure¹⁹ ($126 \pm 32 \text{ kJ mol}^{-1}$), and the value of the diffusivity agrees well with that extrapolated to 3,000 K from lower temperature, low-pressure experiments¹⁹ ($2.4 \times 10^{-9} \text{ m}^2 \text{ s}^{-1}$), indicating that proton diffusion does not depend strongly on melt composition. Error bars represent one-s.d. uncertainties.

- Neutrally buoyant?
Hydrous melt lenses.
Seismic observations (Song et al., 2004).
- Conduction dominated by hydrogen diffusion.
- P=14 GPa, T=1800 K:
 $\sigma \sim 18 \text{ S/m}$.
- Conduction should be high enough to be detectable, through electromagnetic sounding.

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Environment Geochemistry

Berkeley Pit; Butte, Montana
Mining operation: 1955 - .
Copper, silver, gold.

2007: 1249 Superfund-sites in the USA

Toxic metals and formation of SO₄²⁻.

Objectives:

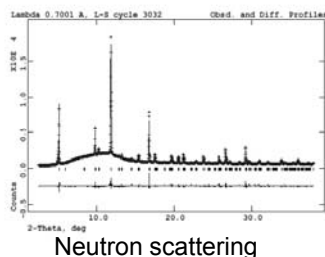
- Restoration of mining areas.
- Water management.
- Environmental hazards.
- Ecology.
- Superfund sites.

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Crystallography of Complex Structured Sulfate Minerals - $(\text{H}_3\text{O})^+\text{Fe}(\text{SO}_4)_2$

(Majzlan and Kiefer, in prep.)

Objective:
Crystal Structure =
Theory + Neutrons + X-rays?



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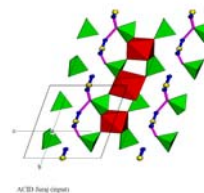
Problem:

• Hydrogen positions cannot be determined reliably → structure only partially solved.

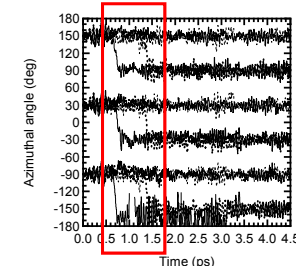
Idea: Heavy cations from experiment and hydrogen positions from theory.

Ab-Initio Structure Refinement

$(\text{H}_3\text{O})^+\text{Fe}(\text{SO}_4)_2$



Initial Structure:
Heavy ions neutron scattering; H planar group



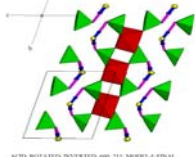
After ~1-1.5 ps rotations of half the H_3O^+ ions.

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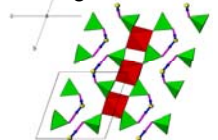
Majzlan and Kiefer (in prep.)

Structure Evaluation

Final structure:
Ab-initio MD



Degenerate State



$\Delta E \approx 0.07 \text{ meV / atom}$

Activation energy:
~4.1 meV/atom
(300 K == 25 meV)

- Groundstate of $(\text{H}_3\text{O})^+\text{Fe}(\text{SO}_4)_2$: superposition of two states.
- Dynamical Disorder.
- May explain difficulty of hydrogen refinements.

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Summary

- Classical Potentials: Pro's and Con's.
- Density-Functional-Theory (DFT):
 - Universal, E_{XC} .
 - Pseudopotentials.
 - Planewaves.
 - Predictive power.
 - Complementary to experiment.
- Finite temperature *ab-initio* Molecular Dynamics.

Theory addresses different classes of problems:

- Determination of significant fact.
- Matching of facts with theory.
- Articulation of theory.

(Categorization after: T. Kuhn, 1962)

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Other Links

[Units and fundamental constants:](http://physics.nist.gov/cuu/Constants/Table/allascii.txt)
<http://physics.nist.gov/cuu/Constants/Table/allascii.txt>